

The Crystal Structure of $\text{Ru}_4\text{Al}_{13}$

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The crystal structure of $\text{Ru}_4\text{Al}_{13}$ has been determined and refined by least-squares techniques on the basis of three-dimensional X-ray single-crystal data. The lattice parameters for the monoclinic cell are:

$$a = 15.86 \text{ \AA}, b = 8.188 \text{ \AA}, c = 12.74 \text{ \AA}, \beta = 107.8^\circ$$

The phase is isotypic with $\text{Fe}_4\text{Al}_{13}$, space group $C2/m$. The relations between $\text{Os}_4\text{Al}_{13}$ and the $\text{Fe}_4\text{Al}_{13}$ -type of structure are discussed.

In previous reports on the crystal chemistry of aluminium alloys of the group VIII metals, the structures of some phases with the composition $\text{Me}_4\text{Al}_{13}$ have been described *viz.* FeAl_3 (of ideal composition $\text{Fe}_4\text{Al}_{13}$)^{1,2} $\text{Co}_4\text{Al}_{13}$ ³ and $\text{Os}_4\text{Al}_{13}$.⁴ The structures of $\text{Fe}_4\text{Al}_{13}$ and $\text{Co}_4\text{Al}_{13}$ are almost identical but differ considerably from that of $\text{Os}_4\text{Al}_{13}$. The compound $\text{Ru}_4\text{Al}_{13}$ described in this work is isomorphous with $\text{Fe}_4\text{Al}_{13}$.

EXPERIMENTAL

An alloy of the composition $\text{Ru}_4\text{Al}_{13}$ was prepared from ruthenium powder (L. Light & Co., about 99.99 %) and aluminium ribbon (E. Merck AG, at least 99.99 %) by melting in an electric arc furnace under an atmosphere of argon. The melt formed by the alloy is very brittle and crystalline. The crystals are prismatic and almost always twinned like those found in the corresponding $\text{Fe}_4\text{Al}_{13}$ compound.¹ A single crystal could, however, be found and this was used to collect complete X-ray single-crystal data with $\text{CuK}\alpha$ radiation. The Weissenberg photographs which were taken with [001] as the rotation axis showed monoclinic symmetry. About a thousand independent reflections from eight layer lines were estimated visually with a standard scale. The crystal was less than 0.01 mm in all dimensions and no correction for absorption was applied. The refinement of the structure was made by using a least-squares program on the computer FACIT EDB.⁵

THE REFINEMENT OF THE STRUCTURE

The structural similarity between the present phase and $\text{Fe}_4\text{Al}_{13}$ was obvious from the X-ray data. Guinier powder photographs, registered with

Table 1. Weight analysis in Ru₄Al₁₃.

sin θ interval	wA^2	Number of independent reflections	$ F_o $ interval	wA^2	Number of independent reflections
0.00—0.46	0.88	180	0—45	1.16	181
0.46—0.58	0.78	139	45—90	0.98	183
0.58—0.66	0.67	119	90—135	1.36	228
0.66—0.73	1.45	112	135—180	0.76	140
0.73—0.79	1.18	88	180—225	0.93	98
0.79—0.84	1.04	98	225—270	0.72	56
0.84—0.88	0.80	73	270—315	0.49	32
0.88—0.92	1.11	77	315—360	0.33	21
0.92—0.96	0.76	46	360—405	0.59	15
0.96—0.99	1.70	51	405 <	0.51	33

CuK α_1 radiation and using KCl as an internal standard, gave the following cell dimensions in Å

$$\begin{aligned}
 a &= 15.862 \pm 0.006 \\
 b &= 8.188 \pm 0.003 \\
 c &= 12.736 \pm 0.004
 \end{aligned}
 \qquad
 \beta = 107.77^\circ \pm 0.08$$

These dimensions are similar to those reported for Fe₄Al₁₃ and Co₄Al₁₃.

Fe₄Al₁₃ has been described by Black^{1,2} as having the symmetry *C2/m*. The structure of Co₄Al₁₃ described by Hudd and Taylor³ is very similar to

Table 2. Final atomic parameters and their e.s.d.'s.

Atom	x	$\sigma(x)$	y	$\sigma(y)$	z	$\sigma(z)$	$B \text{ \AA}^2$	$\sigma(B) \text{ \AA}^2$
Ru ₁	0.0857	0.0002	0		0.3827	0.0003	0.41	0.04
Ru ₂	0.4017	0.0002	0		0.6233	0.0003	0.52	0.04
Ru ₃	0.0920	0.0002	0		0.9885	0.0003	0.51	0.04
Ru ₄	0.4007	0.0002	0		0.9853	0.0003	0.56	0.04
Ru ₅	0.3193	0.0001	0.2961	0.0002	0.2781	0.0002	0.57	0.03
Al ₆	0.0667	0.0007	0		0.1740	0.0012	1.05	0.18
Al ₇	0.3208	0.0007	0		0.2812	0.0013	1.13	0.18
Al ₈	0.2383	0.0007	0		0.5354	0.0013	1.30	0.19
Al ₉	0.0724	0.0006	0		0.5805	0.0012	0.79	0.17
Al ₁₀	0.2396	0.0007	0		0.9565	0.0013	1.43	0.20
Al ₁₁	0.4805	0.0006	0		0.8277	0.0011	0.55	0.15
Al ₁₂	$\frac{1}{2}$	0	0		$\frac{1}{2}$		1.09	0.25
Al ₁₃	0.3112	0.0006	0		0.7729	0.0011	0.53	0.15
Al ₁₄	0.0834	0.0007	0		0.7874	0.0013	1.33	0.19
Al ₁₅	0.1859	0.0005	0.2197	0.0009	0.1098	0.0009	0.99	0.12
Al ₁₆	0.3657	0.0004	0.2168	0.0008	0.1107	0.0008	0.81	0.12
Al ₁₇	0.1770	0.0005	0.2222	0.0010	0.3342	0.0009	1.41	0.13
Al ₁₈	0.4909	0.0005	*0.2332	0.0009	0.3298	0.0009	1.00	0.12
Al ₁₉	0.3639	0.0004	0.2256	0.0009	0.4822	0.0008	0.88	0.12
Al ₂₀	0		0.2499	0.0012	0		0.69	0.16

* corresponding coordinate given by Black is probably a misprint.

Table 3. Part of the Guinier powder pattern of Ru_4Al_{13} ($CuK\alpha_1$).

$h k l$	$10^5 \sin^2 \theta_{obs}$	$10^5 \sin^2 \theta_{calc}$	I_{obs}	I_{calc}
0 0 1	—	403	—	0.3
$\bar{2}$ 0 0	—	1040	—	0.4
$\bar{2}$ 0 1	—	1048	—	0.0
$\bar{1}$ 1 0	1144	1145	vw	2.8
$\bar{1}$ 1 1	1348	1350	w	13.7
0 0 2	—	1613	—	1.3
1 1 1	1747	1746	w	8.8
$\bar{2}$ 0 1	1839	1839	vwv	1.8
$\bar{2}$ 0 2	1859	1863	vw	2.4
$\bar{1}$ 1 2	—	2363	—	0.1
$\bar{3}$ 1 1	—	3035	—	0.0
1 1 2	—	3154	—	0.5
3 1 0	—	3225	—	0.2
2 0 2	3442	3444	st	51.1
$\bar{2}$ 0 3	3482	3484	st	56.2
0 2 0	3438	3540	m	19.2
0 0 3	3626	3630	st	79.8
$\bar{3}$ 1 2	—	3652	—	0.2
$\bar{4}$ 0 1	3773	3773	m	20.2
0 2 1	3946	3943	vw	5.8
4 0 0	4164	4160	st	43.9
$\bar{1}$ 1 3	—	4181	—	0.1
$\bar{4}$ 0 2	4194	4192	st	41.9
3 1 1	—	4221	—	0.2
2 2 0	~4582	{4580	vst	{56.5
$\bar{2}$ 2 1	—	{4588	—	{42.5
3 1 3	—	5076	—	0.0
0 2 2	5154	5153	st	62.8
4 0 1	—	{5354	—	{19.8
1 1 3	—	{5368	—	{0.6
2 2 1	~5380	{5378	vst	{35.4
$\bar{2}$ 2 2	—	{5402	—	{26.9
4 0 3	—	{5418	—	{51.2
2 0 3	5854	5856	vw	8.9
$\bar{2}$ 0 4	5912	5912	vw	10.8
3 1 2	—	6025	—	0.4
0 0 4	6458	6453	vw	4.5
$\bar{5}$ 1 1	—	6800	—	0.0
$\bar{1}$ 1 4	—	6807	—	0.4
2 2 2	6982	6984	vw	4.4
$\bar{5}$ 1 2	—	7022	—	0.0
$\bar{2}$ 2 3	7026	7024	vw	3.3
0 2 3	1775	7170	vw	6.9

Fe_4Al_{13} but was found to have the lower space group Cm . The latter symmetry permits the separation of 8- and 4-fold positions of $C2/m$ into two sets of 4- and 2-fold positions respectively.

The investigation of Ru_4Al_{13} was started using the structure reported for Fe_4Al_{13} as a trial model. The structure was refined by the least-squares method. The weights were calculated using Hughes' weighting scheme. The final R value was 0.114. The weight analysis obtained in the last cycle of refinement is given in Table 1, the final atomic parameters in Table 2 and the Guinier pattern in Table 3.

Table 4. The interatomic distances in Ru₄Al₁₃. Upper limit 3.3 Å. The numbers represent: central atom, neighbour (number of such neighbours), followed by the distance in Å.

1— 6(1) 2.582	6— 1(1) 2.582	12— 2(2) 2.528	17— 1(1) 2.517
8(1) 2.602	3(1) 2.514	18(4) 2.859	2(1) 2.727
9(1) 2.591	3(1) 2.727	19(4) 2.798	5(1) 2.638
9(1) 2.687	14(1) 2.573		6(1) 2.891
14(1) 2.887	15(2) 2.902	13— 2(1) 2.712	7(1) 3.148
17(2) 2.517	17(2) 2.891	4(1) 2.644	8(1) 3.049
18(2) 2.621	20(2) 2.964	5(2) 2.583	8(1) 2.894
19(2) 2.795		8(1) 2.890	13(1) 2.688
	7— 5(2) 2.425	10(1) 2.895	15(1) 2.904
2— 8(1) 2.490	15(2) 3.119	11(1) 2.559	18(1) 2.957
11(1) 2.520	16(2) 3.053	15(2) 2.731	19(1) 2.988
12(1) 2.528	17(2) 3.148	17(2) 2.688	19(1) 2.645
13(1) 2.712	18(2) 3.208		
17(2) 2.727	19(2) 3.060	14— 1(1) 2.887	18— 1(1) 2.621
18(2) 2.507		3(1) 2.522	2(1) 2.507
19(2) 2.519	8— 1(1) 2.602	5(2) 2.579	5(1) 2.649
	2(1) 2.489	6(1) 2.573	7(1) 3.208
3— 3(1) 3.022	5(2) 3.255	9(1) 2.588	9(1) 2.791
6(1) 2.514	9(1) 2.860	10(1) 2.742	11(1) 2.903
6(1) 2.727	13(1) 2.890	16(2) 2.659	12(1) 2.859
10(1) 2.495	17(2) 3.049	18(2) 2.705	14(1) 2.705
14(1) 2.522	17(2) 2.894		16(1) 2.892
15(2) 2.538	19(2) 2.943	15— 3(1) 2.538	17(1) 2.957
16(2) 2.817	19(2) 2.739	4(1) 2.758	19(1) 3.197
20(2) 2.543		5(1) 2.586	19(1) 2.771
	9— 1(1) 2.591	6(1) 2.902	
4— 4(1) 3.056	1(1) 2.687	7(1) 3.119	19— 1(1) 2.795
10(1) 2.469	5(2) 2.664	10(1) 2.965	2(1) 2.519
11(1) 2.679	8(1) 2.860	10(1) 2.826	5(1) 2.543
11(1) 2.547	9(1) 2.569	13(1) 2.731	7(1) 3.060
13(1) 2.644	14(1) 2.588	16(1) 2.848	8(1) 2.943
15(2) 2.758	18(2) 2.791	16(1) 2.725	8(1) 2.739
16(2) 2.560	19(2) 2.683	17(1) 2.904	9(1) 2.683
20(2) 2.555		20(1) 2.863	12(1) 2.798
	10— 3(1) 2.495		17(1) 2.988
5— 7(1) 2.425	4(1) 2.468	16— 3(1) 2.817	17(1) 2.645
8(1) 3.255	5(2) 3.298	4(1) 2.560	18(1) 3.120
9(1) 2.664	13(1) 2.895	5(1) 2.543	18(1) 2.771
10(1) 3.298	14(1) 2.742	7(1) 3.053	
13(1) 2.583	15(2) 2.965	10(1) 2.935	20— 4(2) 2.544
14(1) 2.579	15(2) 2.828	10(1) 2.833	5(2) 2.555
15(1) 2.586	16(2) 2.935	11(1) 2.925	7(2) 2.963
16(1) 2.543	16(2) 2.833	14(1) 2.659	11(2) 2.948
17(1) 2.638		15(1) 2.848	15(2) 2.863
18(1) 2.649	11— 2(1) 2.520	15(1) 2.725	16(2) 2.904
19(1) 2.543	4(1) 2.679	18(1) 2.892	
	4(1) 2.547	20(1) 2.904	
	13(1) 2.559		
	16(2) 2.925		
	18(2) 2.903		
	20(2) 2.948		

Table 5. Distances in $\text{Me}_4\text{Al}_{13}$ compounds and their elements.

Compound	Bond	CN	Range of distance in Å	Average distance in Å
Al fcc	Al—Al	12		2.86
Fe bcc 20°C	Fe—Fe	8		2.48
Fe fcc 916°C	Fe—Fe	12		2.57
Ru hep	Ru—Ru	6 + 6	2.65—2.71	2.68
Os hep	Os—Os	6 + 6	2.68—2.74	2.71
$\text{Fe}_4\text{Al}_{13}$	Fe—Al	~10	2.26—2.84	2.55
	Al—Al		2.45—	
$\text{Ru}_4\text{Al}_{13}$	Ru—Al	~10	2.43—2.89	2.60
	Al—Al		2.56—	
$\text{Os}_4\text{Al}_{13}$	Os—Al	10, 11	2.46—2.86	2.65
	Al—Al	~10	2.57—2.96	2.79

The non-centrosymmetric $\text{Co}_4\text{Al}_{13}$ was then taken as a trial structure in a second calculation but the result showed no indications of $\text{Ru}_4\text{Al}_{13}$ having a lower symmetry than $C2/m$.

Several aluminium sites of $\text{Fe}_4\text{Al}_{13}$ and $\text{Co}_4\text{Al}_{13}$ are reported to be partially occupied by 30—70 % aluminium. According to the B values given in Table 2 there seems to be no such extensive partial occupation by aluminium in $\text{Ru}_4\text{Al}_{13}$.

DISCUSSION

The interatomic distances are given in Table 4. The standard deviations for these are less than 0.02 Å. The numbering of the atoms in this table and in Table 2 is the same as used by Black in the first paper on $\text{Fe}_4\text{Al}_{13}$.¹ In the second paper on $\text{Fe}_4\text{Al}_{13}$,² where the interatomic distances are given, the atoms Al_7 and Al_{14} have been interchanged.

A comparison between the atomic distances of $\text{Fe}_4\text{Al}_{13}$ and $\text{Ru}_4\text{Al}_{13}$ shows that the coordinations around equivalent atoms in the two structures are the same. However, the ranges of distance are different as shown in Table 5. This is because of the extremely short distances found in $\text{Fe}_4\text{Al}_{13}$. On the other hand, the ranges of distance found in $\text{Ru}_4\text{Al}_{13}$ are almost identical to those found in $\text{Os}_4\text{Al}_{13}$, except for the upper limit of the Al—Al distances (*cf.* Table 5). According to the distribution of Al—Al distances around 3 Å found in $\text{Ru}_4\text{Al}_{13}$ there is no definite upper limit like that found for Al—Al distances in $\text{Os}_4\text{Al}_{13}$. Well defined coordination shells indicating close packing are formed around all the atoms of $\text{Os}_4\text{Al}_{13}$ and all the ruthenium atoms of $\text{Ru}_4\text{Al}_{13}$. This is not the case with all the aluminium atoms of $\text{Ru}_4\text{Al}_{13}$ and is caused by the aluminium atom numbered 7 which has only two contacts closer than 3.0 Å.

The average distances in the elements are given in Table 5. Os and Ru have about the same size and are considerably larger than Fe. According to this and the fact that $\text{Os}_4\text{Al}_{13}$ is better packed, $\text{Ru}_4\text{Al}_{13}$ should have the $\text{Os}_4\text{Al}_{13}$ structure from geometrical reasons. This is, however, not the case and the Ru_5 atom seems to play an important role in this connection. The arrangement around Ru_1 — Ru_4 is very similar to the arrangement around the Os atoms

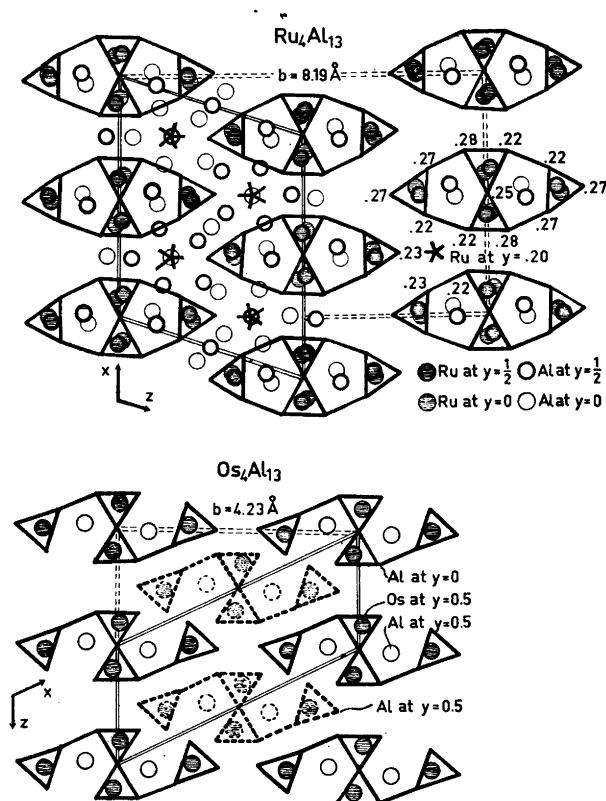


Fig. 1. The atomic arrangements in $\text{Os}_4\text{Al}_{13}$ and $\text{Ru}_4\text{Al}_{13}$. There is an aluminium atom at each corner of the triangles. These aluminium atoms form a puckered net at around $y = 1/4$ in $\text{Ru}_4\text{Al}_{13}$. The deviation from planarity is indicated in one of the figures of $\text{Ru}_4\text{Al}_{13}$. The cell dimensions for $\text{Os}_4\text{Al}_{13}$ are: $a = 17.64 \text{ \AA}$, $b = 4.228 \text{ \AA}$, $c = 7.773 \text{ \AA}$, $\beta = 115.15^\circ$.

in $\text{Os}_4\text{Al}_{13}$ but differs completely from that around Ru_5 . This fact is illustrated in Fig. 1. $\text{Os}_4\text{Al}_{13}$ may be described as built up from column-like arrangements of the composition $\text{Os}_4\text{Al}_{13}$ the projections of which are indicated in the figure. Very similar units of the same composition and containing Ru_1-Ru_4 are found in the ruthenium phase. The Ru_5 atoms, at the asterisk-like positions in the figure, form almost straight $\text{Ru}_5-\text{Al}_7-\text{Ru}_5$ bonds along $[010]$ and force Al, to have no other contact closer than 3.0 \AA .

No explanation why the composition is the same in the two compounds has been found.

When looking at a three-dimensional model of $\text{Ru}_4\text{Al}_{13}$ the most striking feature is the regular arrangement of the Ru atoms. Ru_5 is in the centre of a deformed pentagonal prism formed by the other ruthenium atoms. Only fragments of this pattern are found in $\text{Os}_4\text{Al}_{13}$.

Compounds of similar composition and structures seem to exist in the rhodium and palladium systems. The crystals of these form very complicated twins but give a zero layer Weissenberg photograph around a rotation axis of 16.6 Å, which, as far as the strong reflections are concerned, is very like the corresponding zero layer photograph of $\text{Ru}_4\text{Al}_{13}$. The strong spots are approximately arranged in a 5-fold symmetry but this is more exactly so in the Pd and Rh compounds and these have much larger unit cells.

It seems likely that the approximately five-fold symmetry observed in the single-crystal patterns of these new, very complicated phases reflects an even more regular transition-metal structure than that found in $\text{Ru}_4\text{Al}_{13}$.

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